
Structure and Morphology of Silicon-Germanium Thin Films

**B. G. Demczyk-
Air Force Research Laboratory
SUNY, Stony Brook**

Collaborators:

**D. Bliss – Air Force Research Laboratory
A. King - SUNY, Stony Brook**

Island Formation

- **Significance:**
 - Critical to growth of multilayer stacks
 - Of interest as self-assembling arrays

Island Formation

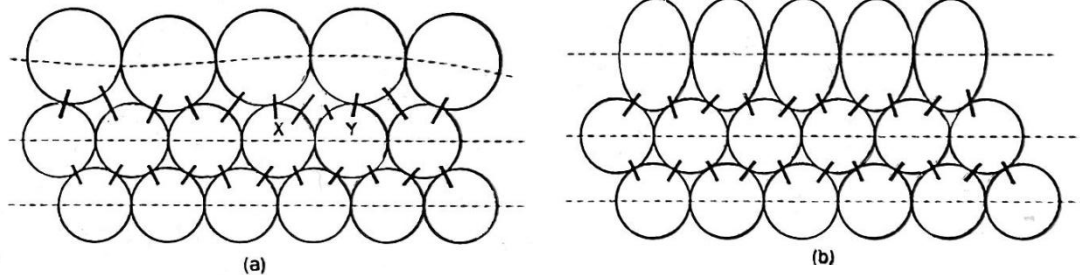
- **Interest Areas:**

- Growth under tensile and compressive strain

- Growth on (Si, Ge) (001) and (111)

- Local lattice deformation in film

SiGe Strained Layer



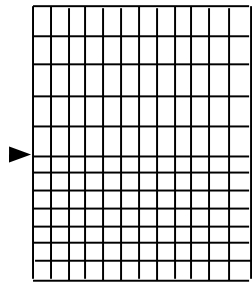
adapted from Reference 1

↓ grows as ↓

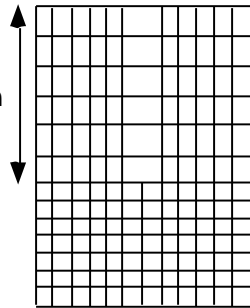
(a) $h < h_c$

(b) $h > h_c$

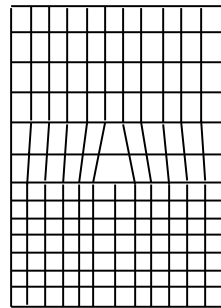
(c) $h > h_c$



Pseudomorphic

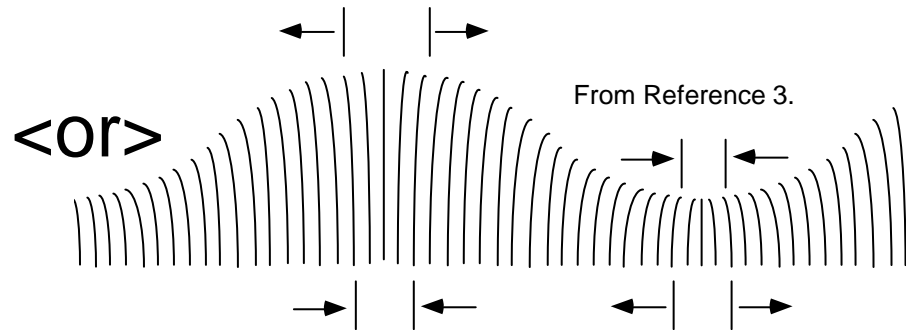


From Reference 2.



Dislocated

<or>



From Reference 3.

Strain-free islands

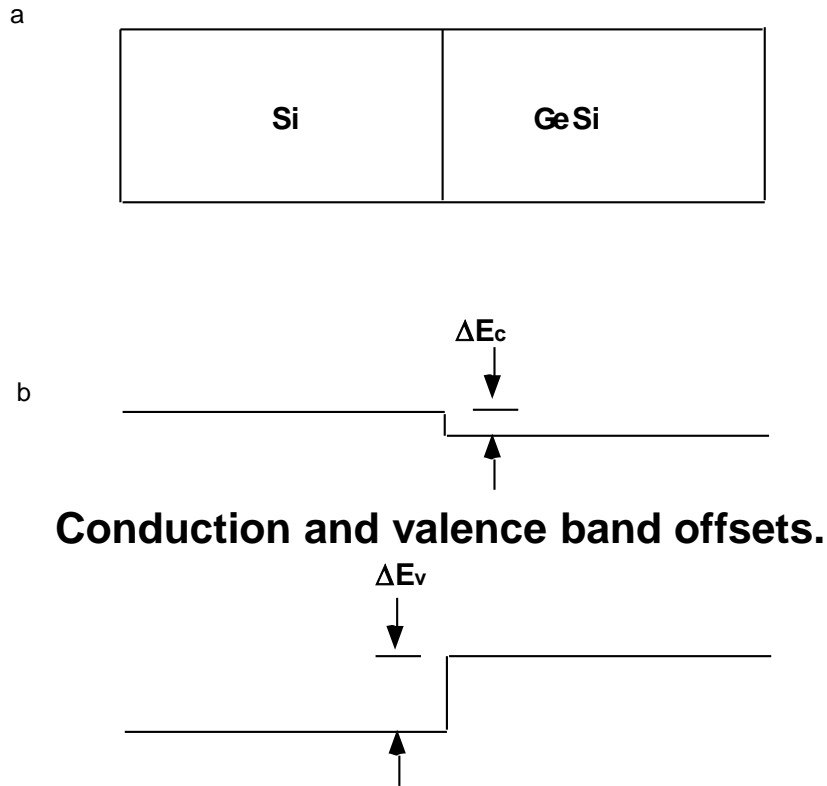
Planar w/dislocations.

h_c = critical thickness for dislocation

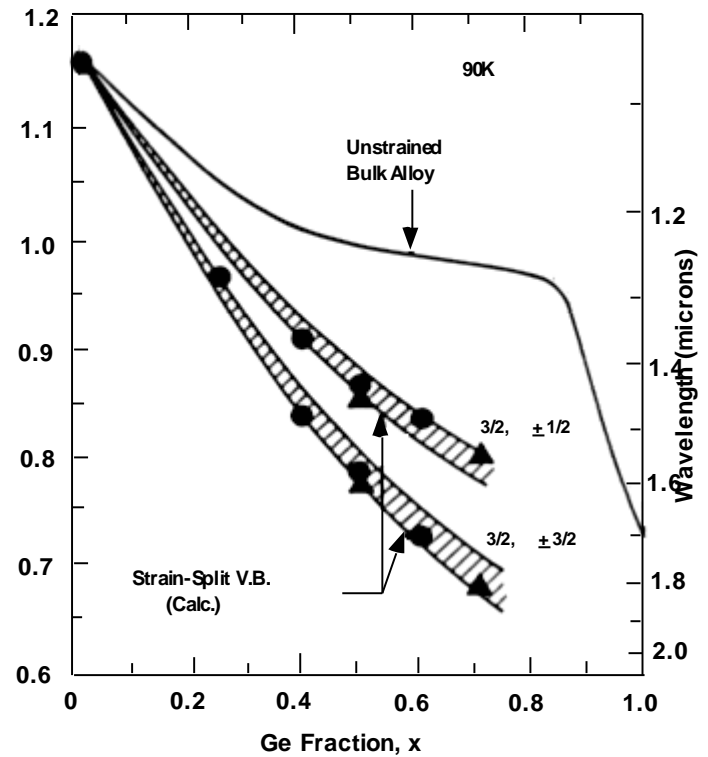
formation $= (b/f)[\ln(hc/b) + 1]$,

(b = Burger's vector)

Band Offset

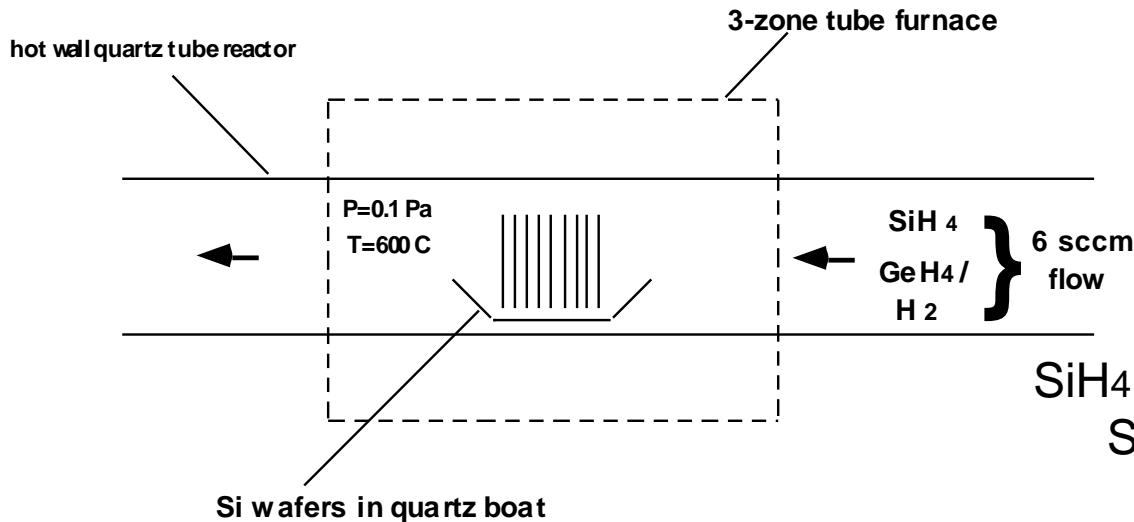


adapted from Reference 4.



From Reference 5.

Thin Film Growth: UHV/CVD



For Silicon:



X = surface site

x* is species, x, bound to surface site

Gas (low pressure) Kinetics:

$$Z_i = N_a / 2\pi M_i k_b T)^{-1/2} p_i$$

Z_i = flux of species, i,

p_i = partial pressure

M_i = molar mass of species, i

N_a = Avogadro's number

k_b = Boltzman's constant

T = absolute temperature

Depositions Made:

substrates: Si and Ge (100) and (111)

compositions: Si, Ge and Si_{0.5}Ge_{0.5}

substrate T_s = 600 C

thickness ~ 100 nm

Alloy Thin Film Synthesis: Kinetics

Fluxes, Z:

$$Z_{\text{SiH}_4} = A(M_{\text{SiH}_4} T)^{-1/2} p_{\text{SiH}_4}$$

$$Z_{\text{GeH}_4} = A(M_{\text{GeH}_4} T)^{-1/2} p_{\text{GeH}_4}$$

N = active site density,

k = decomposition rate constant

M_i = molar mass

p = partial pressure

A = constant

Growth rate, R:

$$R \sim N(k_{\text{Si}} Z_{\text{SiH}_4} + k_{\text{Ge}} Z_{\text{GeH}_4})$$

Germanium fraction, x:

$$x = R_{\text{Ge}} / (R_{\text{Ge}} + R_{\text{Si}}) = [1 + ((M_{\text{GeH}_4} / M_{\text{SiH}_4})^{1/2} (s_{\text{SiH}_4} / s_{\text{GeH}_4}) (p_{\text{SiH}_4} / p_{\text{GeH}_4}))]^{-1}$$
$$= 1 / [(1 + 1.545) (s_{\text{SiH}_4} / s_{\text{GeH}_4}) (p_{\text{SiH}_4} / p_{\text{GeH}_4})]^{-1}$$

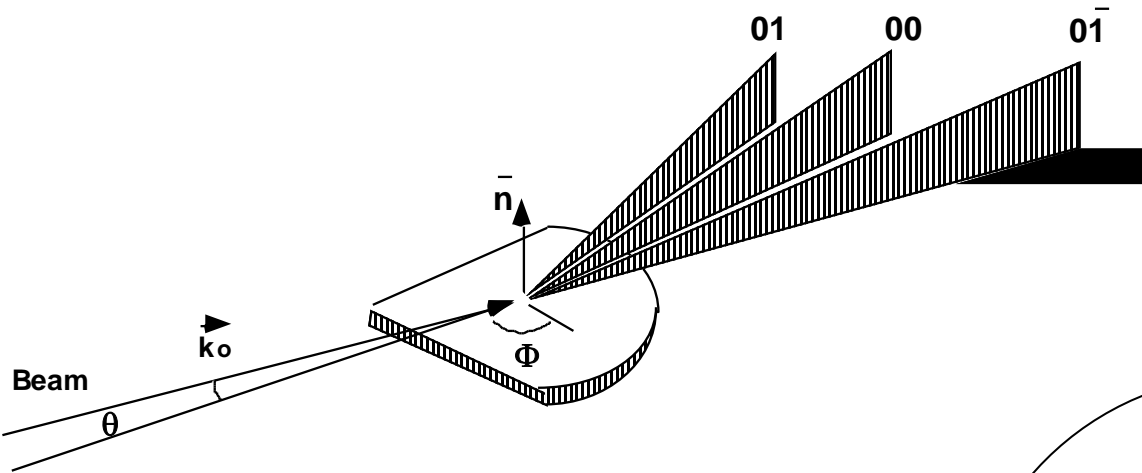
$$1.545 = (M_{\text{GeH}_4} / M_{\text{SiH}_4})^{1/2}$$

empirical, from Reference 4

Technique Summary

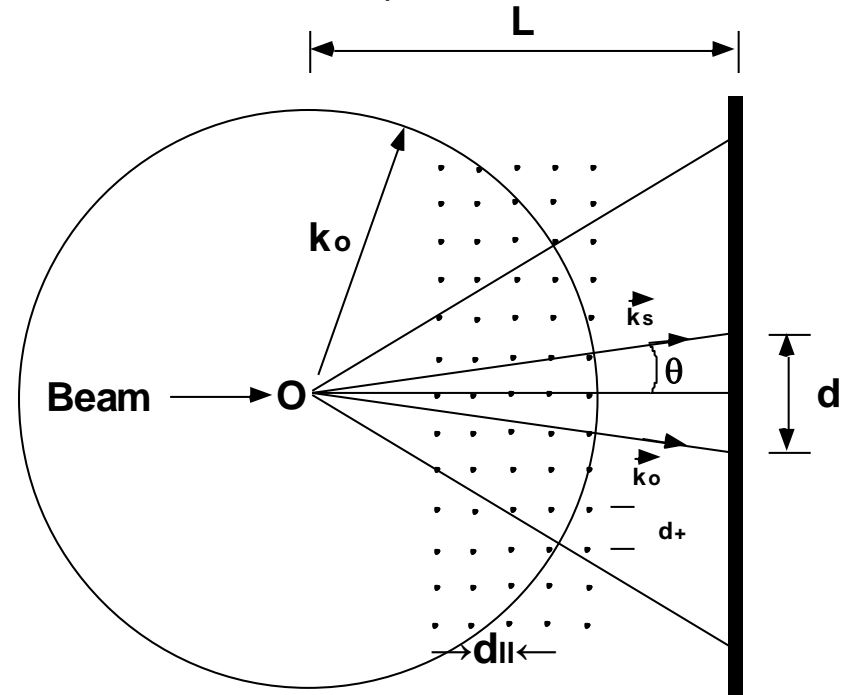
Sampling Regime:	~1 μm	100 nm	1-10 nm
Technique:	AFM	RHEED	EM/HREM
Gauges:	3D asperity size	Crystal "quality"	Character of interface
<i>via:</i>	Surface roughness	In-plane spacing	Local lattice parameters

RHEED Basics



Basic Geometry ↑
adapted from Reference 6

Reciprocal Lattice ↓
adapted from Reference 7

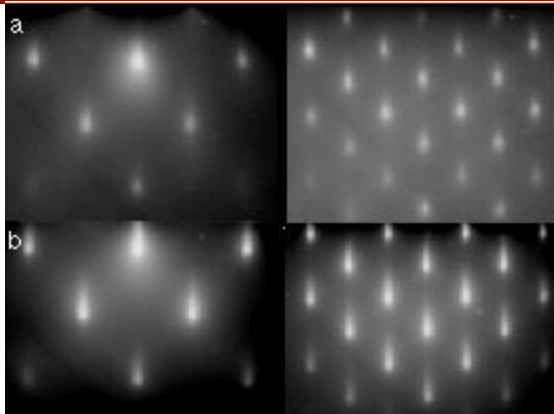


$$d_{||} = \lambda L/d$$

(Si,Ge) Surface- (001)

Ge/Si(001)

→

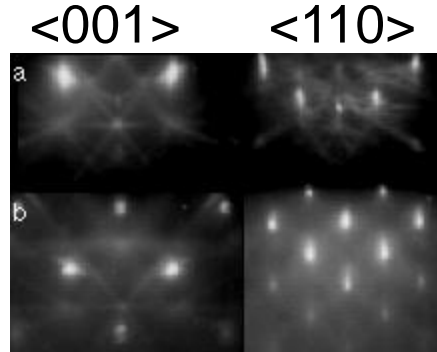


<001> <110>

Si/Si(001)

→

Ge/Ge(001)



- “smoother” (streaks) along <110>

- spacings: <110> “symmetric” $a/4$,

<100> asymmetric $\sqrt{2}/2$

- Si/Si(001): less rough

- Si/Ge(001): <110> elongated spots

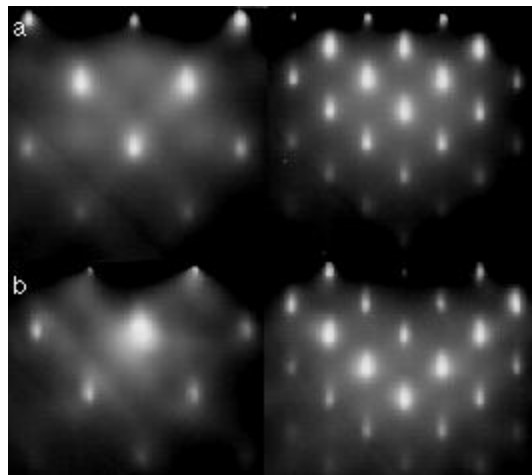
- Ge/Si(001): maxima more discrete than Ge/Ge(001)

- Alloy films: elongated spots

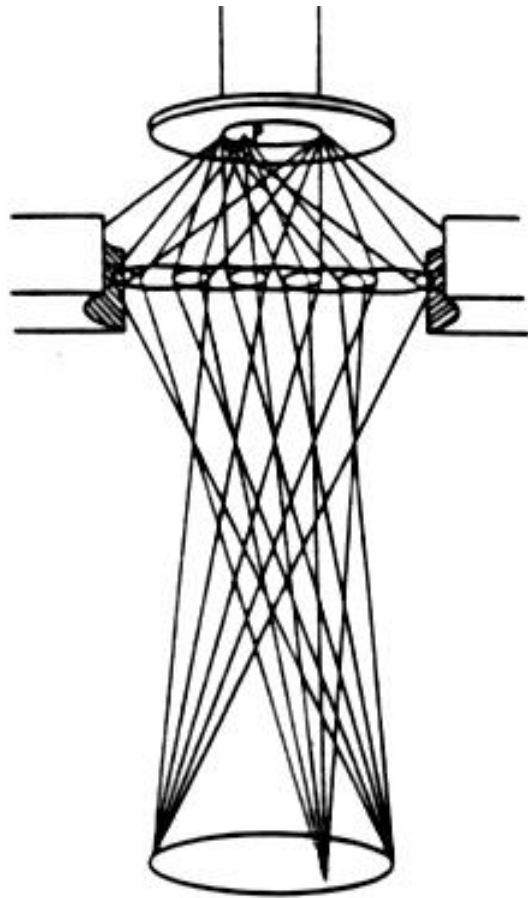
SiGe/Si(001)

→

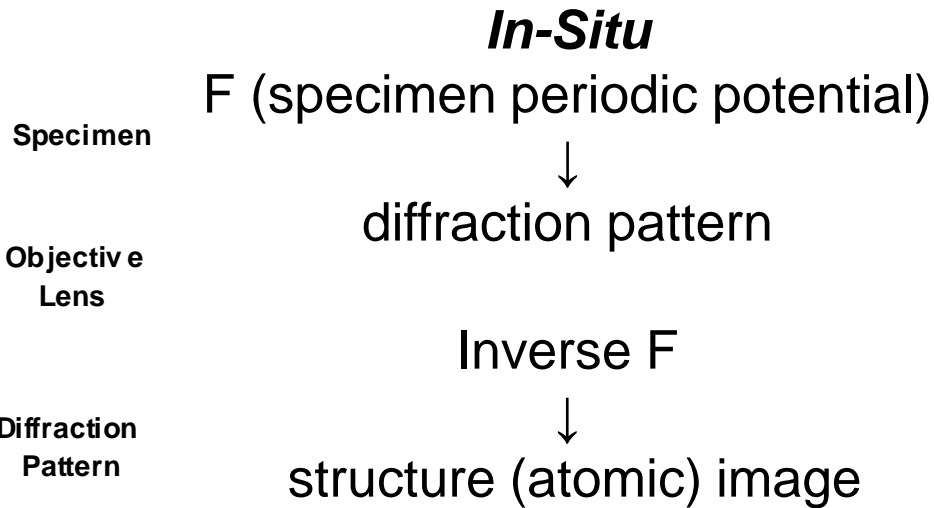
SiGe/Ge(001)



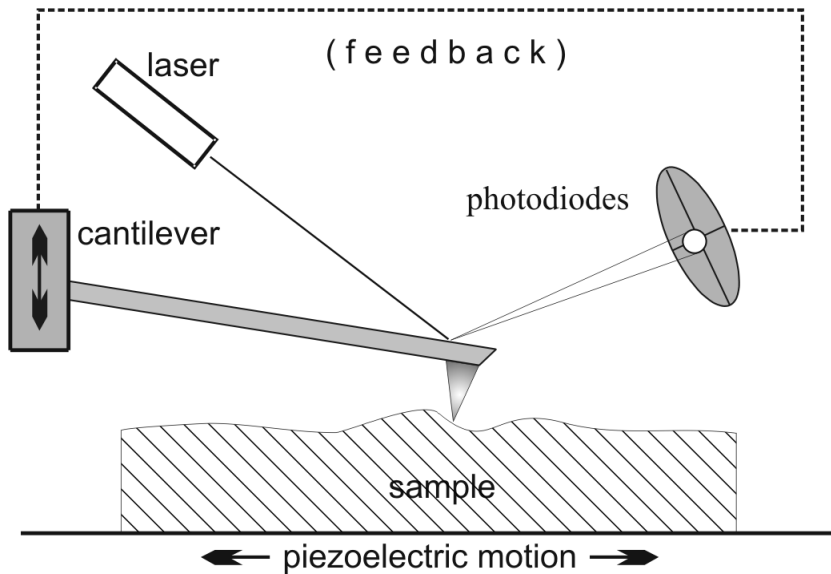
HREM Fundamentals



Schematic Representation
after Reference 8

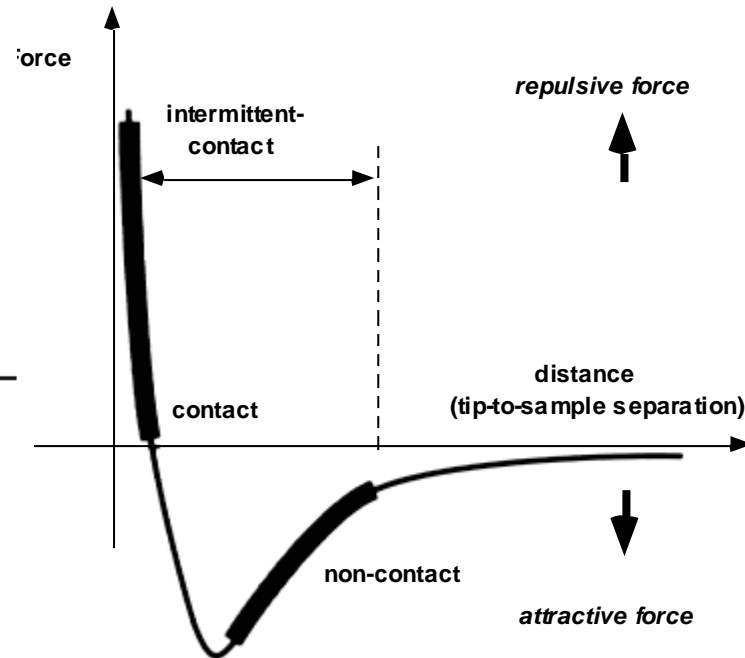


AFM Basics



Schematic Representation

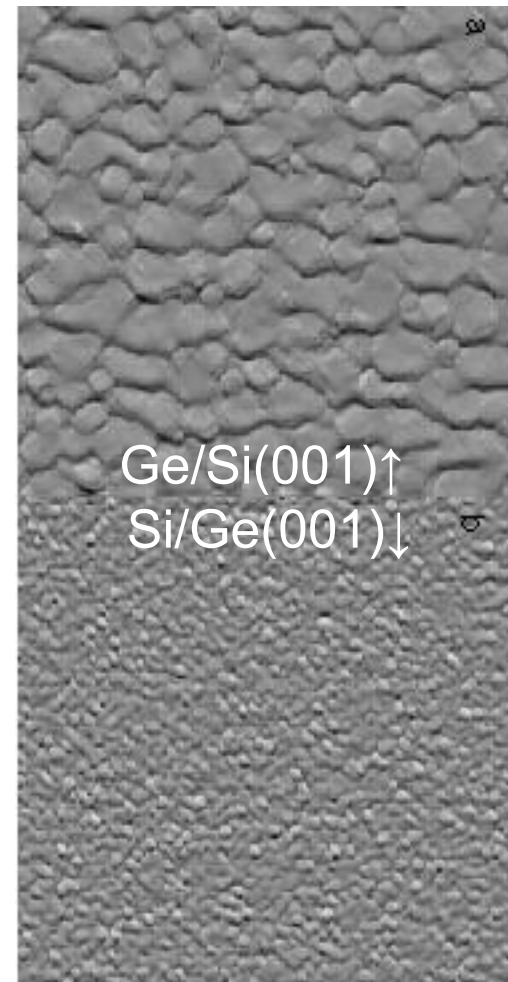
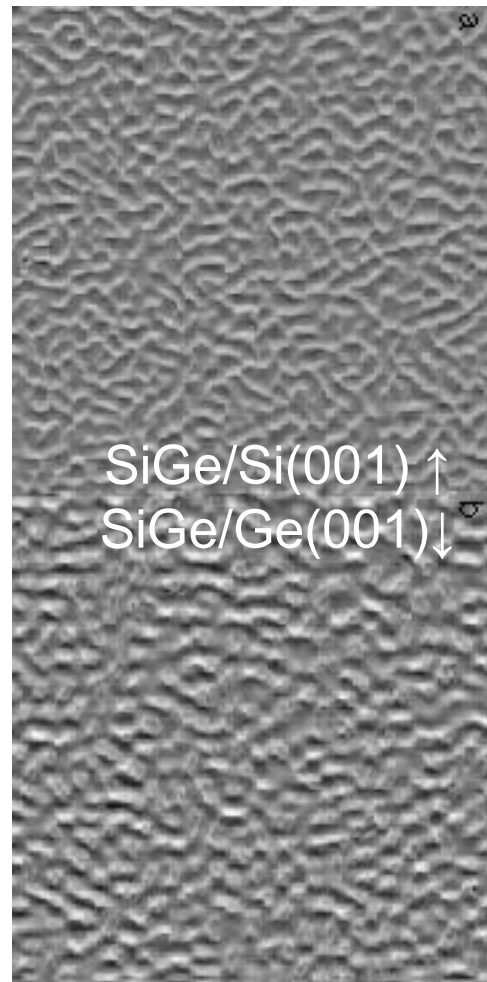
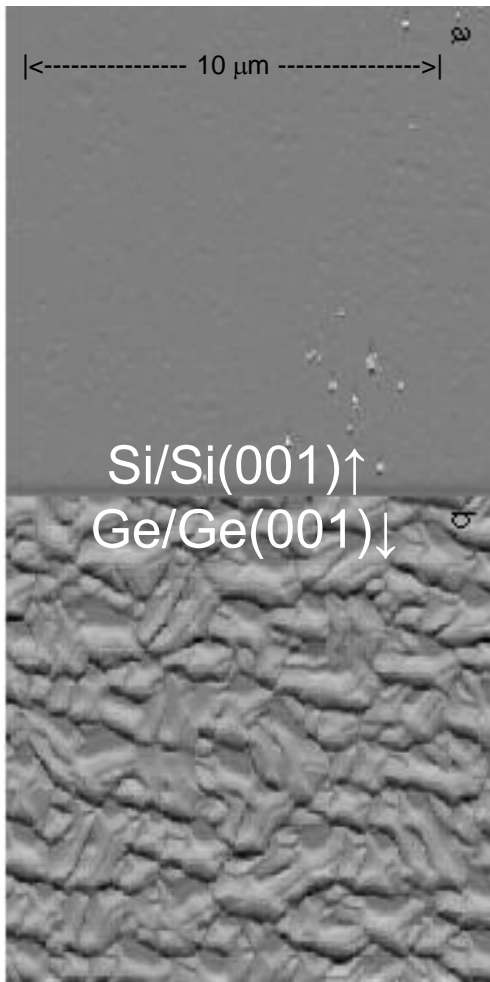
After reference 9



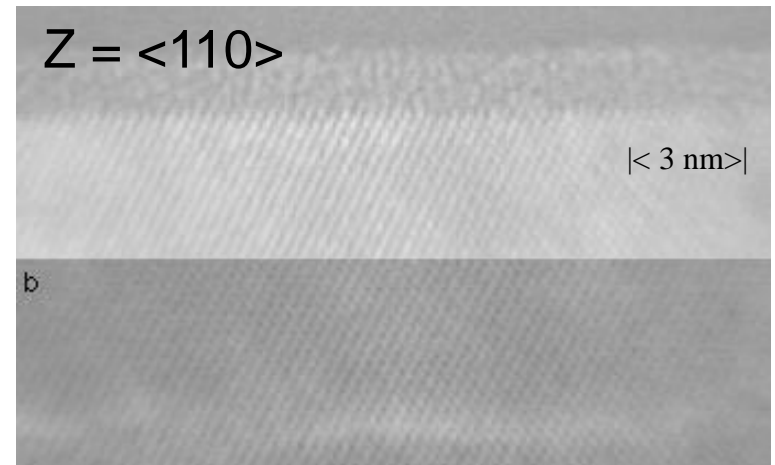
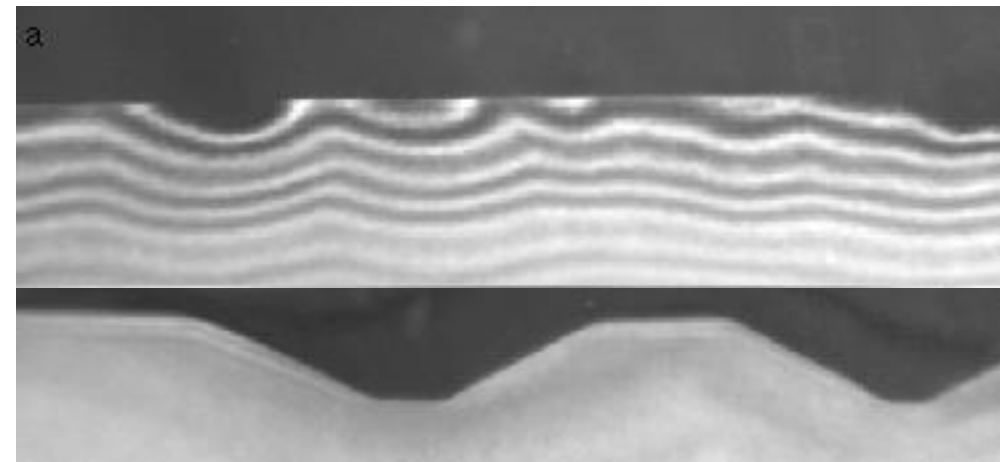
Intermolecular Force vs. Separation

After Reference 10

(Si,Ge) Morphology- (001)



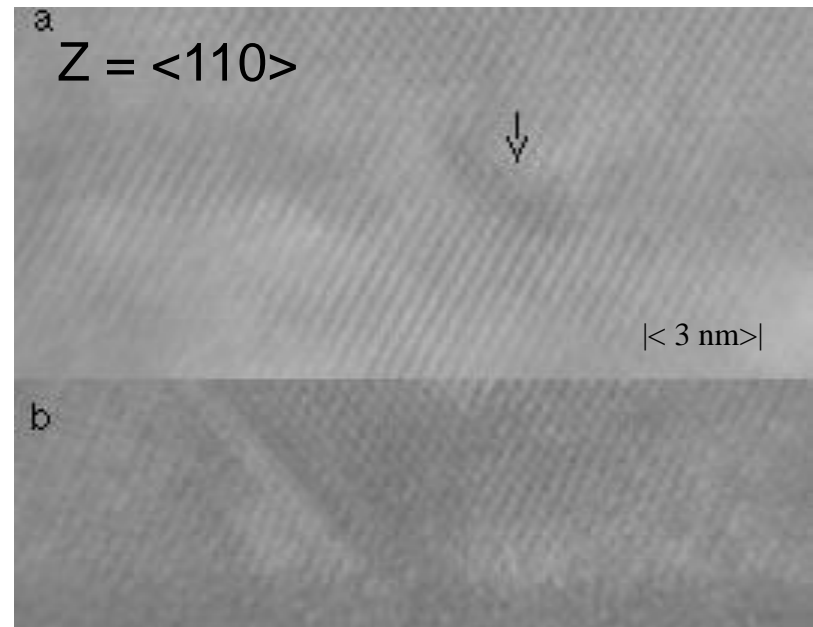
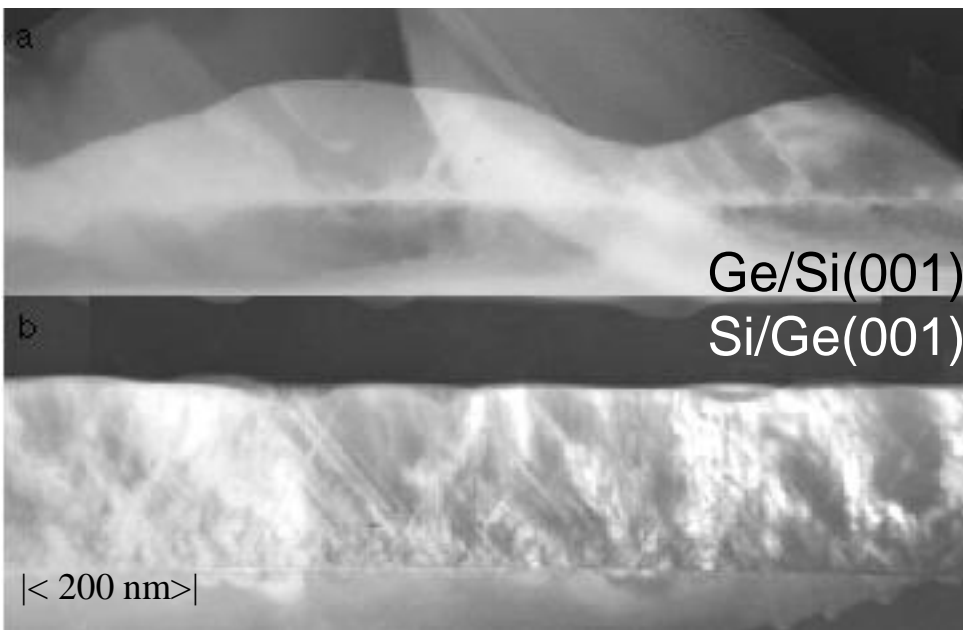
Interface-Zero Strain-(001)



defect-free

Interface-Maximum Strain-(001)

films in compression \rightarrow surface undulations

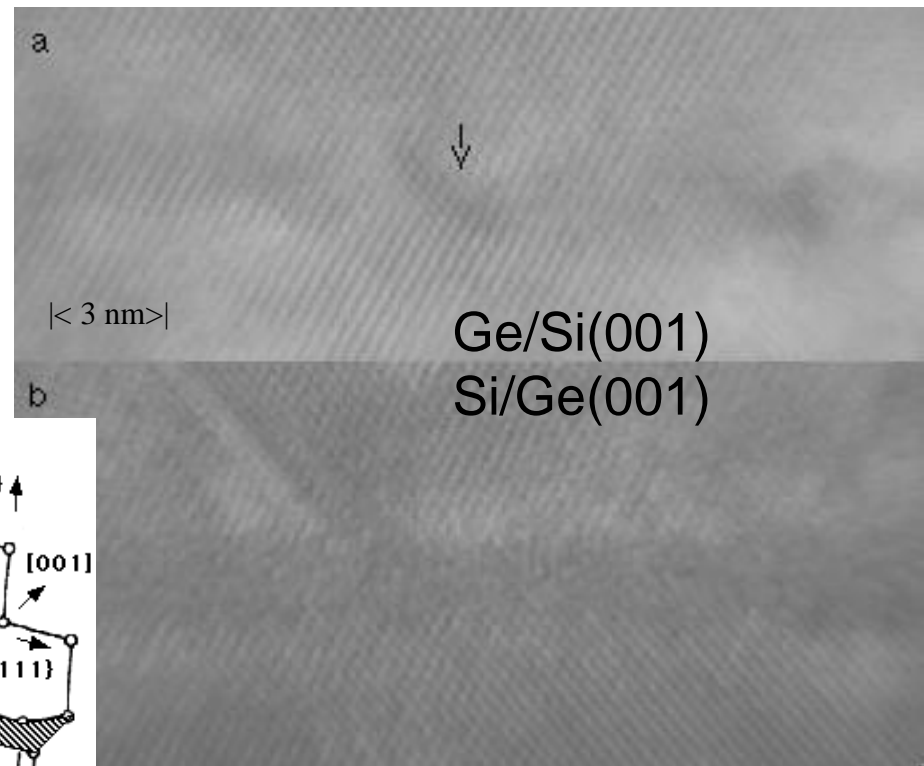
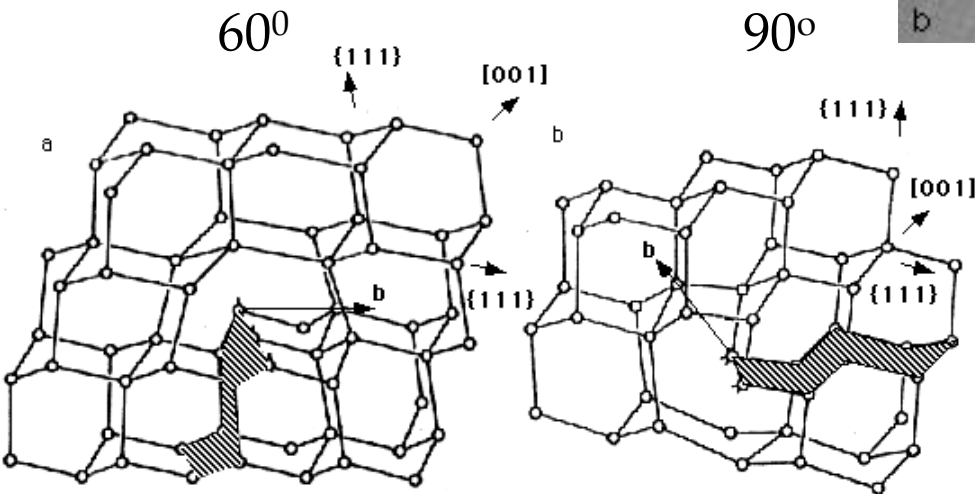


films in tension \rightarrow more defects-

Details of Strained (001) Interface

Roughness related to the presence of dislocation structures.

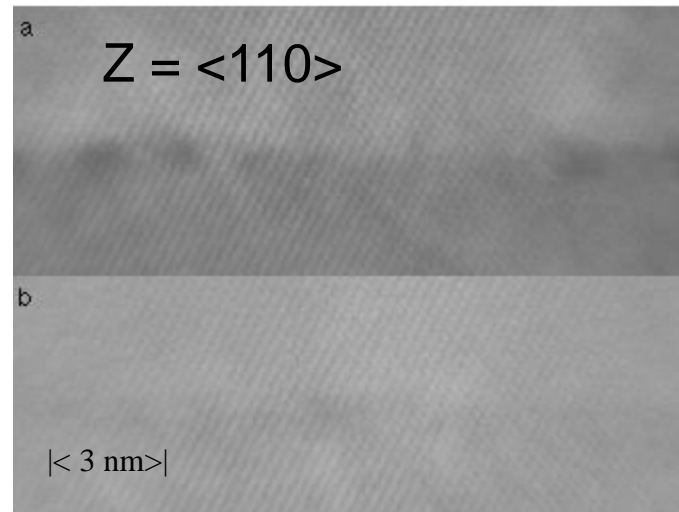
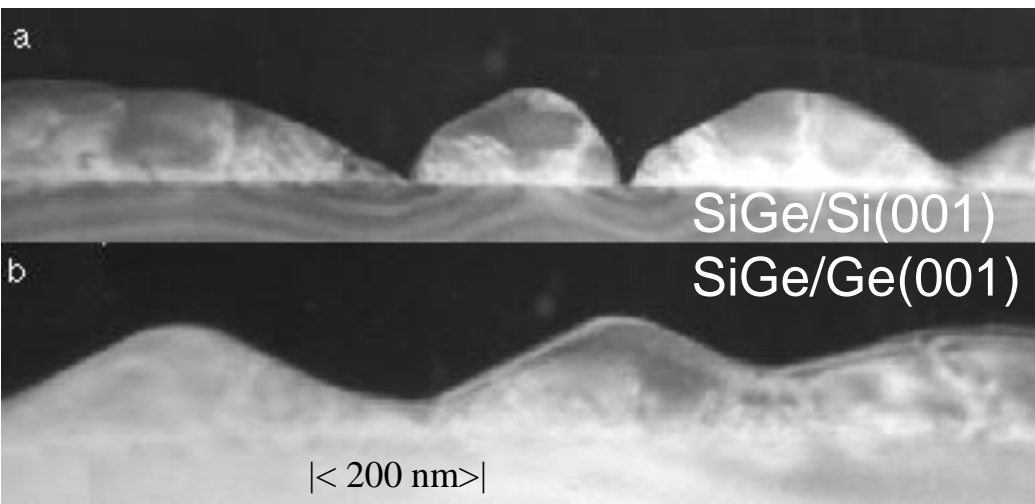
Dislocation structures in
diamond cubic lattice .↓
from Reference 11.



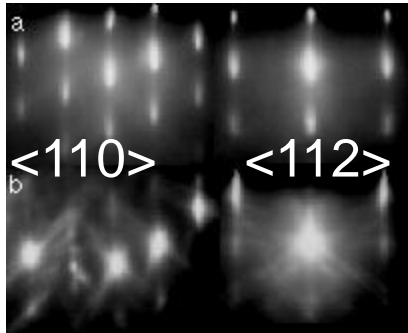
Ge/Si(001)
Si/Ge(001)

Interface-SiGe- (001)

steeply faceted islands



(Si,Ge) Surface-(111)



Si/Si(111)

←

Ge/Ge(111)

Ge/Ge(111): streaked RHEED maxima

Si/Si(111): general surface roughness

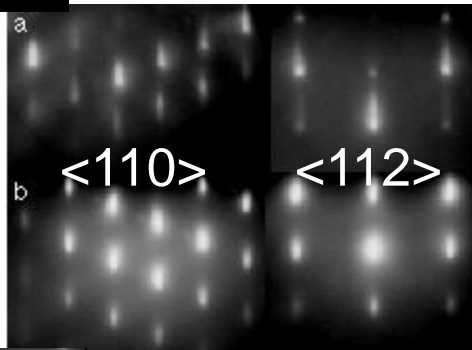
Ge/Ge(111): large scale mesa coverage

- R_a larger since RHEED samples the smooth mesa surface

Ge/Si(111)

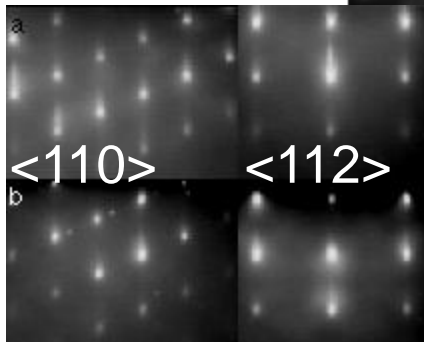
→

Si/Ge(111)



(Ge/Si (111)): smoother than Si/Ge(111)

- Ge/Si(111) consists of large scale plateaus,
- Si/Ge(111) displays general surface roughness



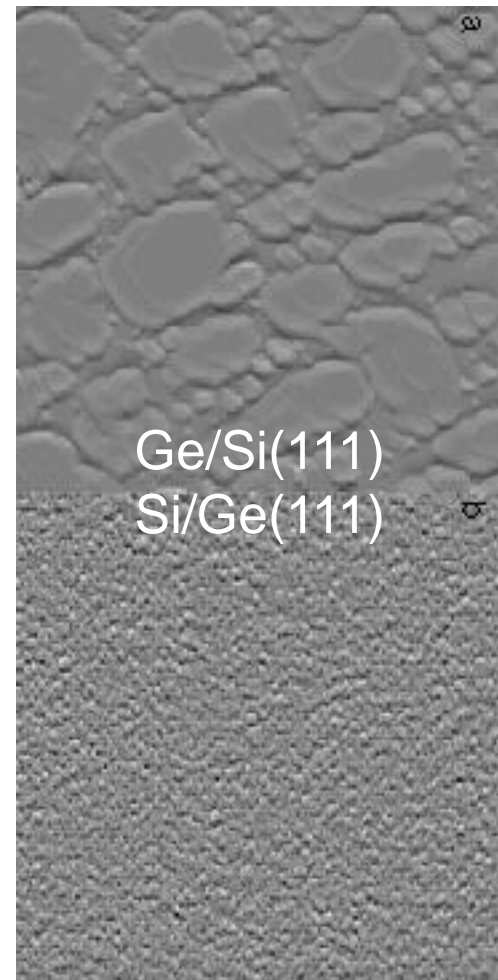
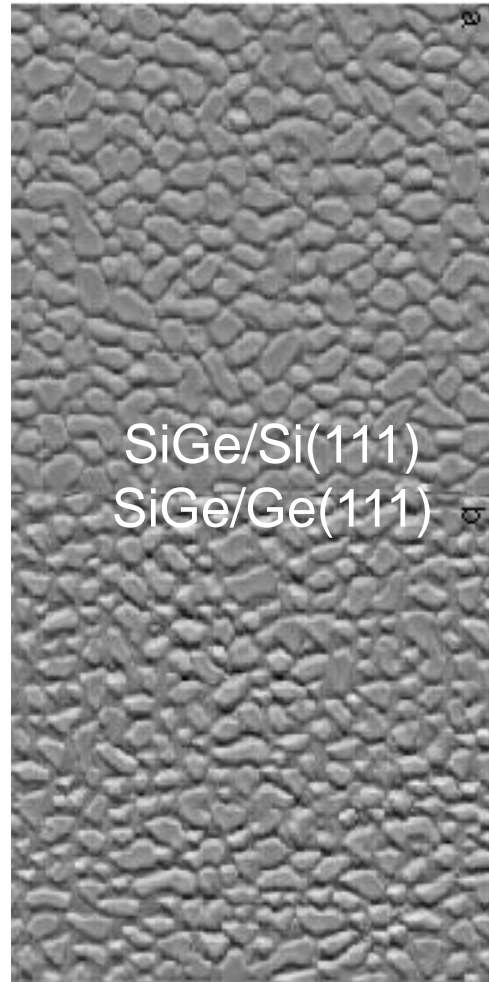
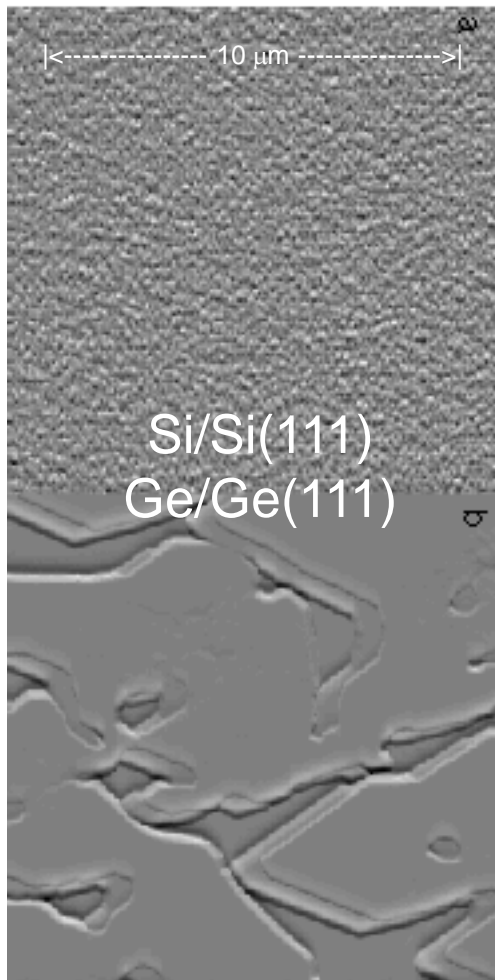
SiGe/Si(111)

←

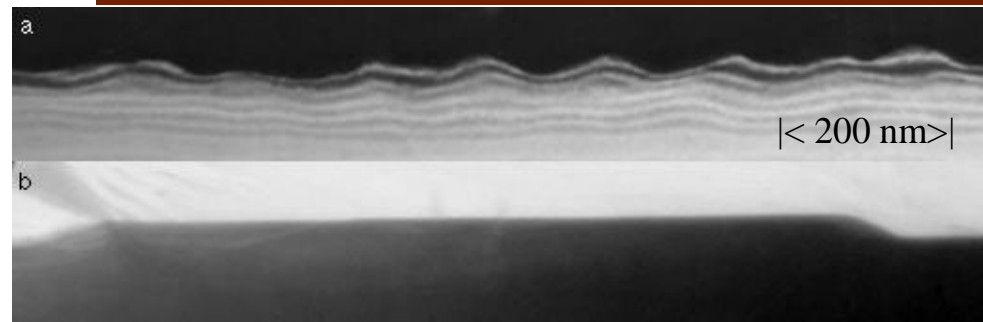
SiGe/Ge(111)

Alloy films on Si (111) and Ge(111) are virtually indistinguishable

(Si,Ge) Morphology–(111)



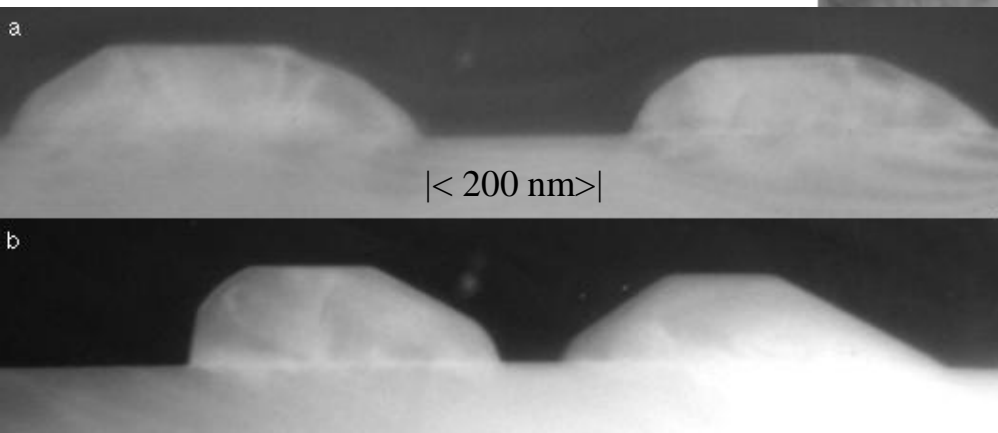
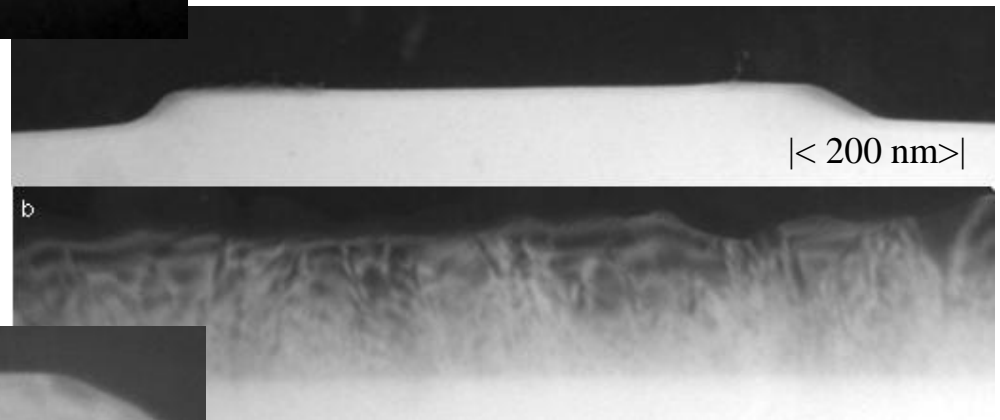
(111) Substrates



Si/Si(111)
Ge/Ge(111)

$Z = \langle 110 \rangle$

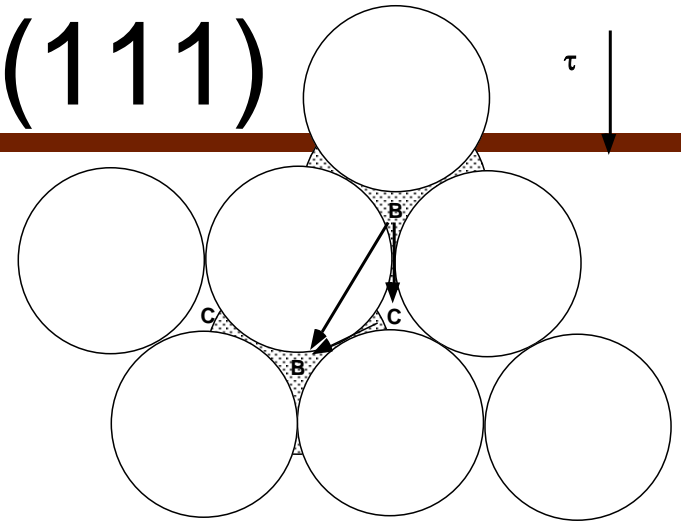
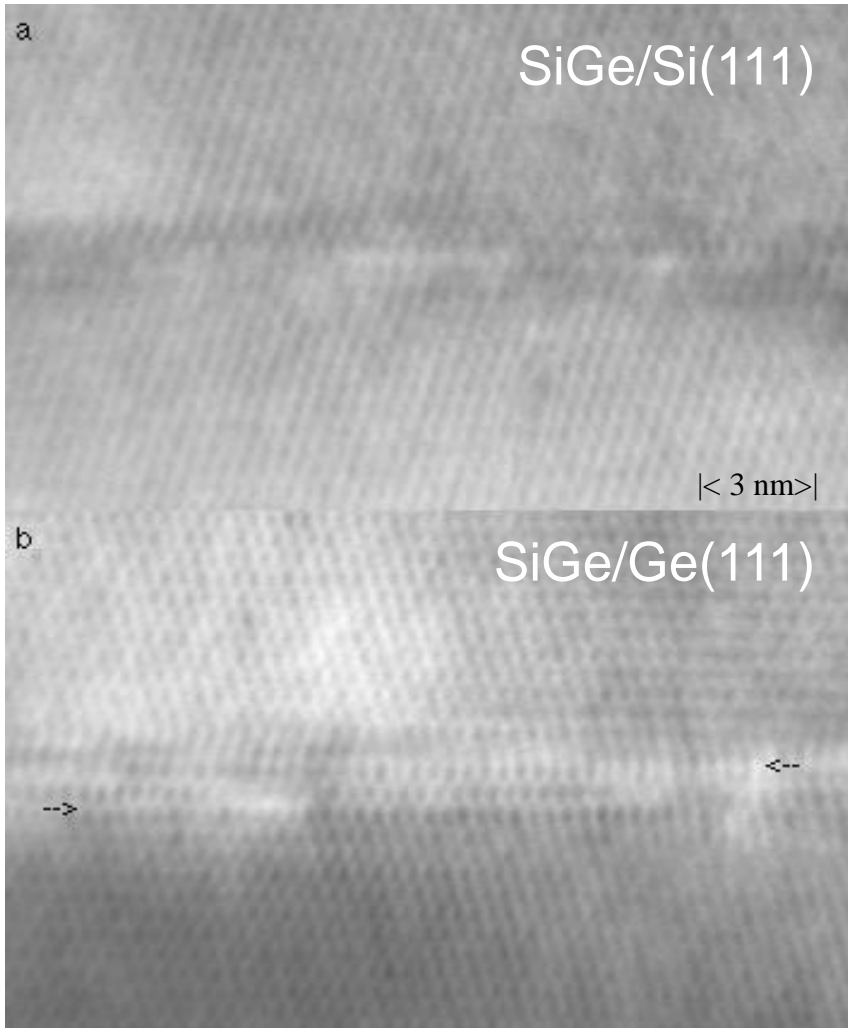
Si/Ge(111)
Ge/Si(111)



SiGe/Si(111)
SiGe/Ge(111)

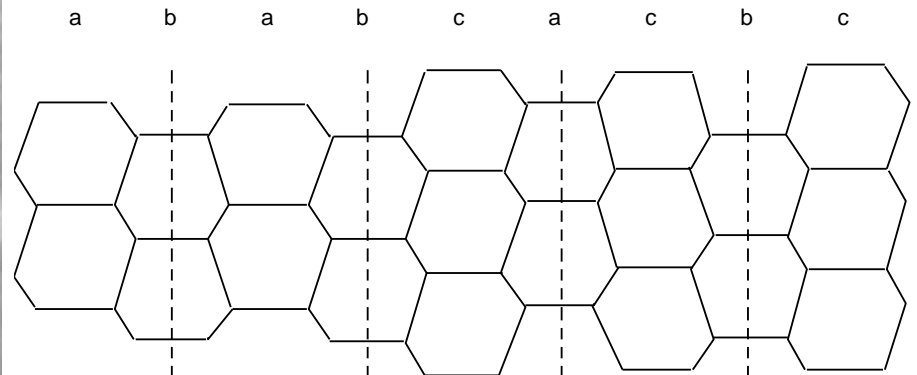
broad plateaus

Interface -SiGe on (111)



Schematic of slip mechanism of 60° dislocation on (111)

From Reference 12



Stacking faults in DC lattice

From Reference 11

HREM Film-Substrate Mismatch of (Si,Ge) Films

**lattice parameter ↑
from substrate to
bulk deposit value
within islands**

	<u>(001) Substrates</u>	<u>(111) Substrates</u>
<u>Zero Strain Samples</u>		
Ge/Ge	.007	
<u>Maximum Strain Samples</u>		
Ge/Si	.045	
Si/Ge	.032	
<u>Alloys</u>		
SiGe/Si	.041	.035
SiGe/Ge	.012	.011

Summary of Observations: Substrate Mismatch of (Si,Ge) Films

Ge/Si(001): higher strain (4.5%) than the Si/Ge(001) (3.2%)
→ tetragonal strain (film constrained to match the substrate along the interface).

Si/Ge(001): high density of defects, which relieve a portion of the 4.5% strain

Alloy films: same mismatch on both (100) and (111)

Larger mismatches for alloys deposited on Si (3.5-4%), than Ge (~1%)
→ evidence for a Ge-rich alloy composition at start of growth (stress-driven diffusion)

Si/Si	Ge/Si (I)	Alloy/Si (I)
Si/Ge (P)	Ge/Ge	Alloy/Ge (I)
Si/Alloy (P)	Ge/Alloy (P)	Alloy/Alloy (I or P)

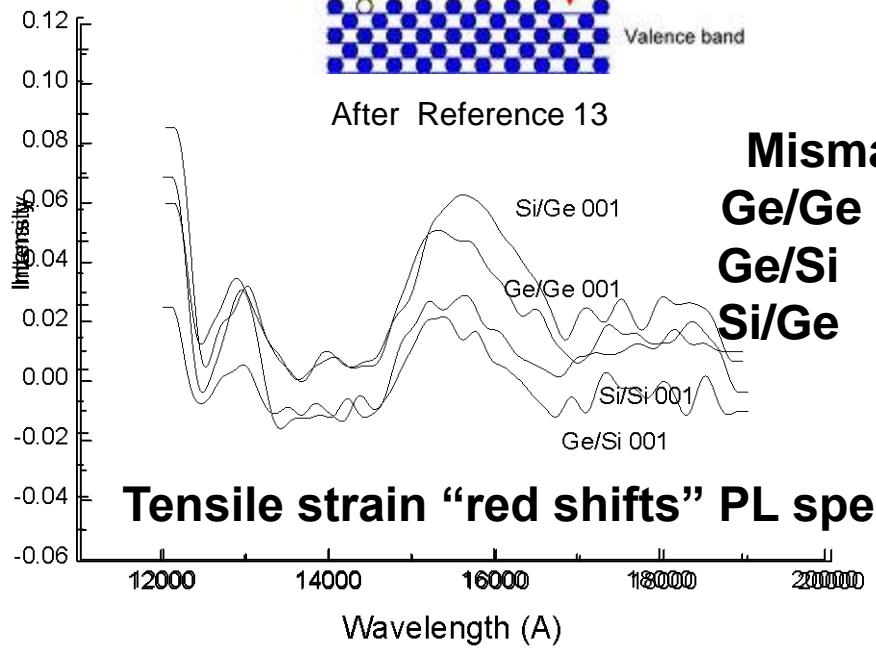
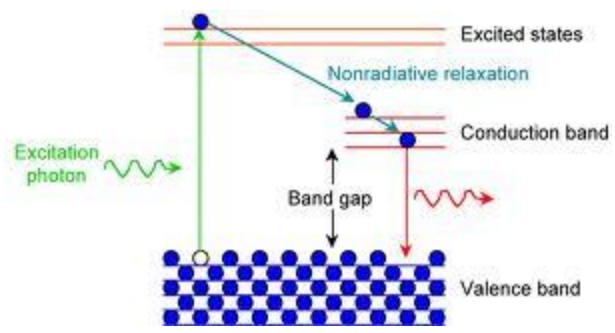
I-island formation
P-planar layer

RHEED Surface Mismatch (e) with Bulk of (Si,Ge) Films

- films adopt equilibrium spacings irrespective of substrate or orientation (completely relaxed)

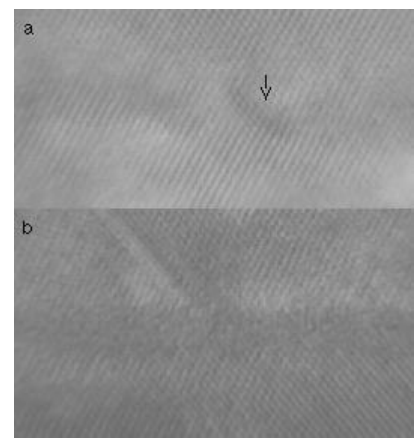
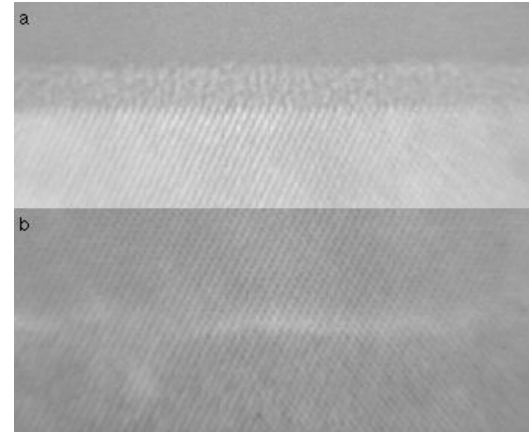
	<u>(001) Substrates</u>		<u>(111) Substrates</u>	
	e_{Si}	e_{Ge}	e_{Si}	e_{Ge}
<u>Zero Strain Samples</u>				
Ge/Ge	----	-.009	----	-.010
<u>Maximum Strain Samples</u>				
Ge/Si	.052	-.007	.036	-.006
Si/Ge	-.012	-.052	-.006	-.047
<u>Alloys</u>				
SiGe/Si	.017	-.024	.036	-.006
SiGe/Ge	.026	-.015	.025	-.016

Photoluminescence



Si/Si(001)
Ge/Ge(001)

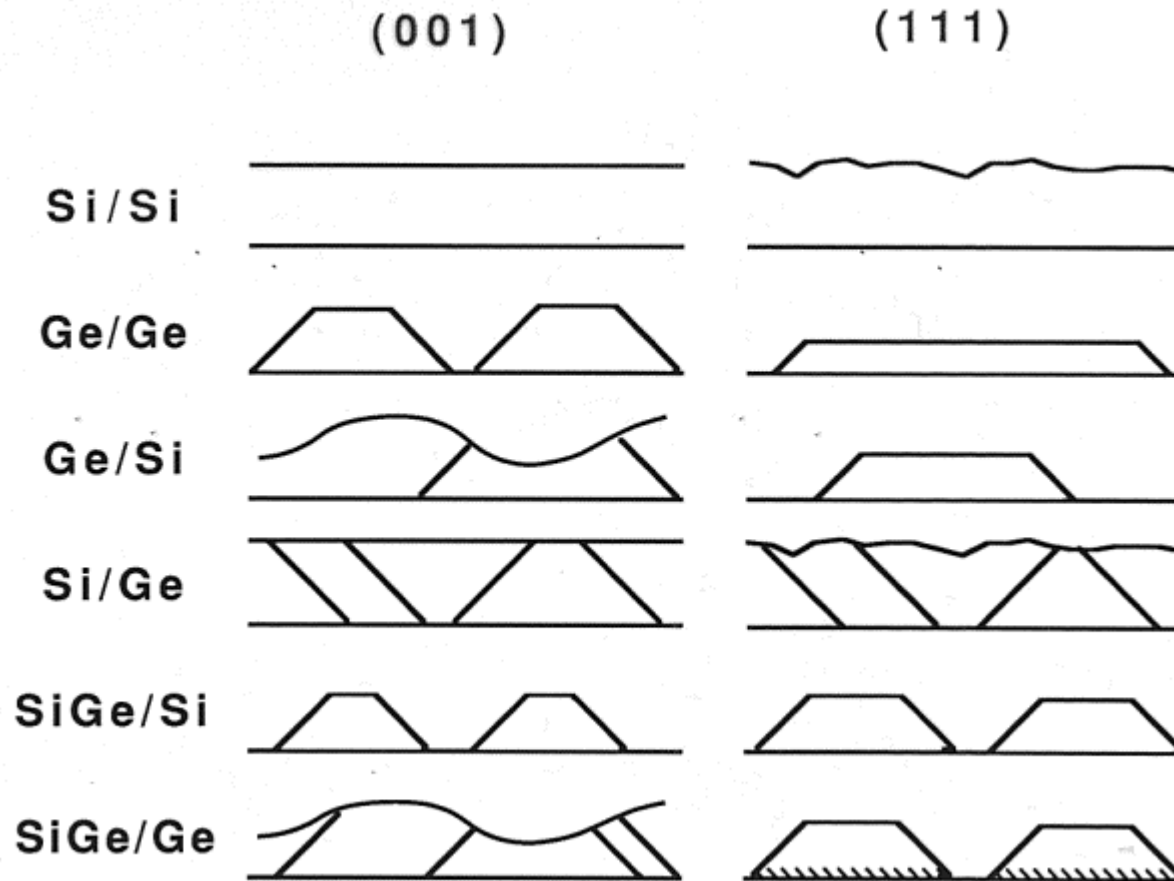
Mismatch:
Ge/Ge .007
Ge/Si .045
Si/Ge .032



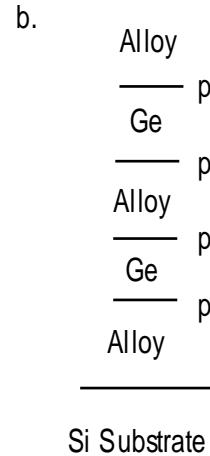
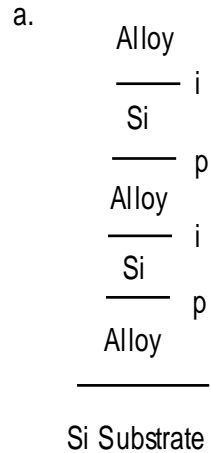
Ge/Si(001)
Si/Ge(001)

The insertion of an extra half plane on the Si side of the interface leaves it in a state of compression and this is manifested in the PL spectrum

Morphologies Observed



Proposed SiGe Multilayer Quantum Dot



Promote dislocation-free, island growth in materials with bandgaps of interest

Via a low σ deposit (Ge or a SiGe alloy) grown on Si (or alloy on Ge), forming islands (σ is surface energy)

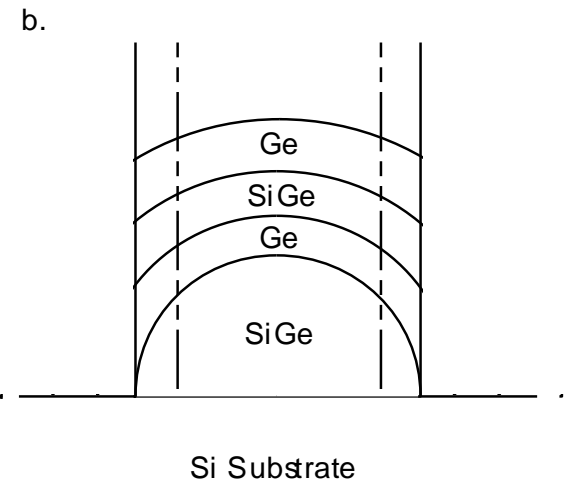
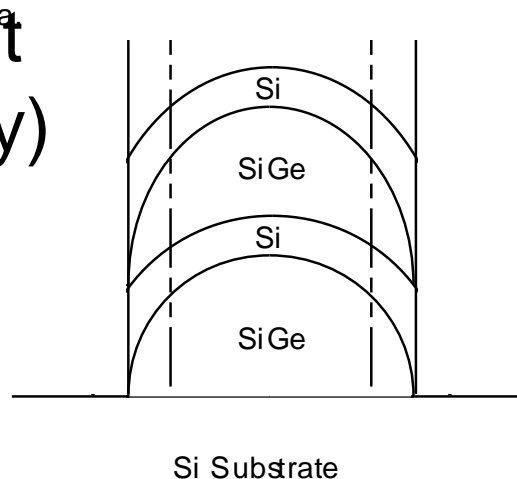


Figure References

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- 6). S. Elagoz, PhD Dissertation, University of Michigan Department of Physics, 1993, p. 24.
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- 11). J. Hornstra, *J. Phys. Chem. Sol.* 5, 129 (1957).
- 12). P. M. Maree, J. C. Barbour, J. F. van der Veen, K. L. Kavanagh, C. W. T. Bulle-Lieuwma and M. P. A. Vieggers, *J. Appl. Phys.*62(11), 4413 (1987).
- 13). http://www.cnx.org/contents/ba27839d-5042-4a40-afcf-c0e6e39fb454@20.16/Physical_Methods_in_Chemistry

Summary-SiGe Growth

- minimization of surface energy controls initial film coverage of substrate
 - *(111) surface area maximized*
 - islands relieve strain
 - *lattice parameters* ↑ *near top of island*
 - *compete with dislocation formation*
 - islands reduce surface energy
 - *form even with zero strain*
 - tensile stress favors dislocations
- 2D to 3D growth transition depends on:
energy difference, misfit & modulus

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AFM – P. Yip

MIT Lincoln Laboratory

HREM – P. Nitishin

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